

# Polymers

(continued)

18.S995 - L14 & 15

## 2.1.2 vMF polymer model

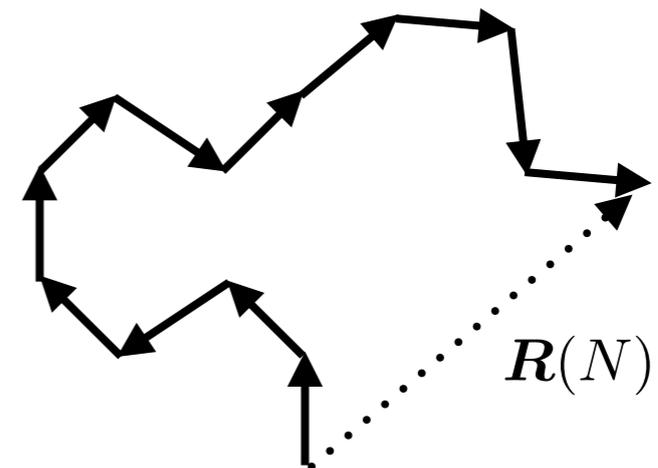
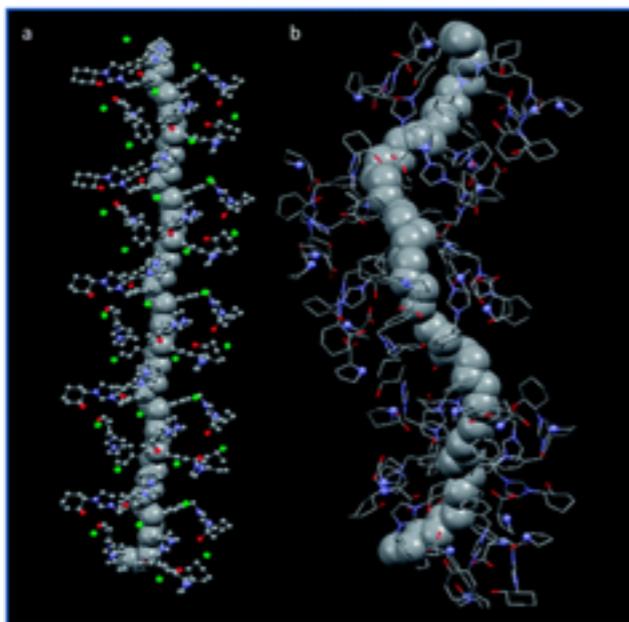
Consider an idealized polymer consisting of  $i = 1, \dots, N$  segments of length  $\lambda$ . Each segment has an orientation  $\boldsymbol{\mu}_i$ , so that the vector connecting the two polymer ends is given by

$$\mathbf{R}(N) = \sum_{i=1}^N \mathbf{R}_i = \lambda \sum_{i=1}^N \boldsymbol{\mu}_i. \quad (2.5)$$

The total length of the polymer is  $L = N\lambda$  and w.l.o.g. we choose  $\mathbf{R}(0)$  and  $\boldsymbol{\mu}_1 = (0, 0, 1)$ . We assume that the conditional PDF of  $\boldsymbol{\mu}_i$  for a given  $\boldsymbol{\mu}_{i-1}$ , is a vMF-distribution with spread parameter  $\kappa$ ,

$$f(\boldsymbol{\mu}_i | \boldsymbol{\mu}_{i-1}) = C_2 e^{\kappa \boldsymbol{\mu}_i \cdot \boldsymbol{\mu}_{i-1}}. \quad (2.6)$$

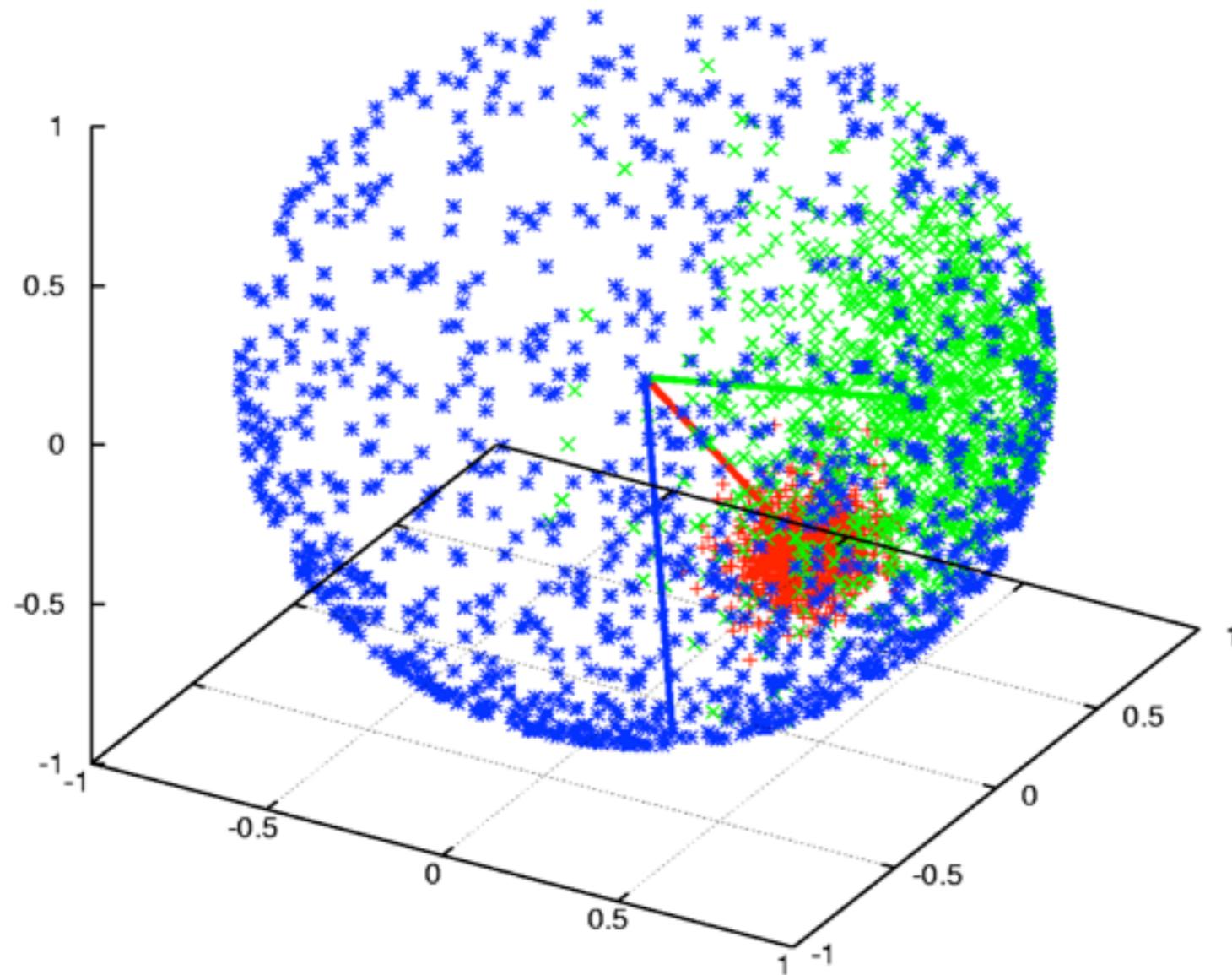
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# von Mises-Fisher distribution



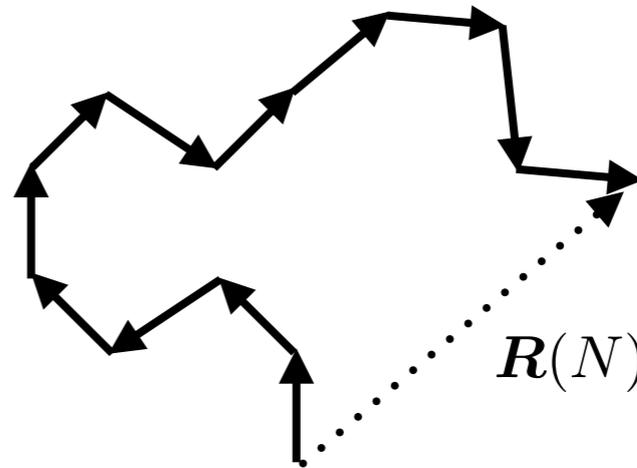
$$\kappa = 1$$

$$\kappa = 10$$

$$\kappa = 100$$

arrows = mean direction

$$\mathbf{R}(N) = \sum_{i=1}^N \mathbf{R}_i = \lambda \sum_{i=1}^N \boldsymbol{\mu}_i.$$



We would like to compute correlation functions and statistical moments of  $\mathbf{R}(N)$  in the limit of large  $N$ . Of particular interest are the mean end-position

$$\mathbb{E}[\mathbf{R}(N)|\boldsymbol{\mu}_1] = \lambda \sum_{n=1}^N \mathbb{E}[\boldsymbol{\mu}_n|\boldsymbol{\mu}_1], \quad (2.7a)$$

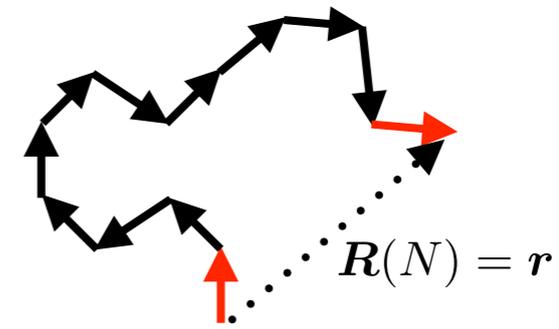
the squared end-to-end distance

$$\mathcal{D}(N) = \mathbb{E}[\mathbf{R}(N) \cdot \mathbf{R}(N)], \quad (2.7b)$$

and the excursion PDF

$$p_N(\mathbf{r}) = \mathbb{E}[\delta(\mathbf{r} - \mathbf{R}(N))]. \quad (2.7c)$$

## Excursion PDF & thermodynamics



$$p_N(\mathbf{r}) = \mathbb{E}[\delta(\mathbf{r} - \mathbf{R}(N))]$$

Unfortunately, it is not possible to compute the excursion PDF (2.7c) exactly for the vMF model<sup>1</sup>. However, the central limit theorem combined with (2.18c) implies that, for large  $N$ , the excursion PDF will approach a Gaussian

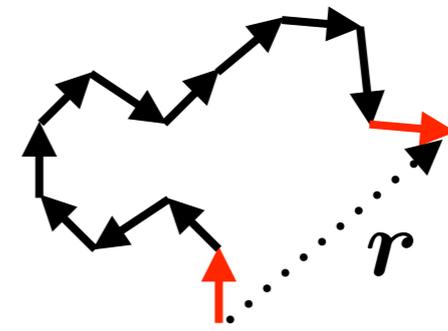
$$p(\mathbf{r}) \simeq \left( \frac{3}{2\pi DN} \right)^{3/2} e^{-3\mathbf{r}^2/(2DN)}. \quad (2.21)$$

For the remainder of this section, we will assume that the end-points of the polymer are fixed at  $\mathbf{0}$  and  $\mathbf{r}$ . To make the connection with thermodynamics, we may consider  $\mathbf{r}$  as a *macroscopic* state-variable, that can be realized by a number of different polymer configurations referred to as *microstates*. If no other constraints are known, it is plausible that each microstate is equally likely and, for large  $N$ , the number of microstates realizing a specific the macrostate  $\mathbf{r}$  is  $\lambda^3 p(\mathbf{r})$ , assuming the spatial resolution is of the order of the segment length  $\lambda$ . The corresponding microcanonical entropy is given by

$$S \simeq k_B \ln[\lambda^3 p(\mathbf{r})] = S_0 - k_B \frac{3\mathbf{r}^2}{2DN}. \quad (2.22)$$

$$D \simeq 2\lambda^2 \kappa = 2\lambda L_P$$

## Excursion PDF & thermodynamics



$$S \simeq k_B \ln[\lambda^3 p(\mathbf{r})] = S_0 - k_B \frac{3r^2}{2DN}$$

To obtain a prediction for the mean force  $\mathbf{f}$  required to stretch the polymer by a small amount  $d\mathbf{r}$ , we can exploit the general thermodynamic relation

$$dE = \delta W + \delta Q, \quad (2.23a)$$

where work and heat increments are defined as usual by

$$\delta W = -\mathbf{f} \cdot d\mathbf{r}, \quad \delta Q = TdS, \quad (2.23b)$$

with  $T$  denoting temperature. If one neglect self-avoidance interactions, which are present in real polymers, the energy remains constant during a change of conformation,  $dE = 0$ . Hence,

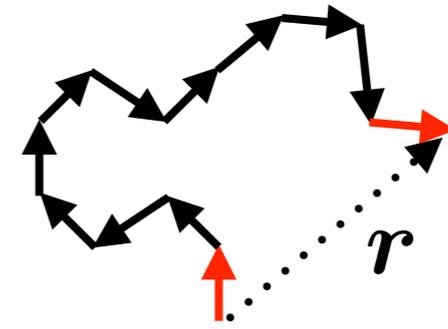
$$dS = \frac{\mathbf{f}}{T} \cdot d\mathbf{r} \quad (2.24)$$

and the stretch force components are obtained as

$$f_i = T \left( \frac{\partial S}{\partial r_i} \right) = -\frac{3k_B T}{DN} r_i. \quad (2.25)$$

$-\mathbf{f}$  is the force needed to stretch a polymer in a solvent bath of temperature  $T$

## Excursion PDF & thermodynamics



$$S \simeq k_B \ln[\lambda^3 p(\mathbf{r})] = S_0 - k_B \frac{3\mathbf{r}^2}{2DN}$$

Furthermore, it is also instructive to compute the corresponding free-energy

$$F := E - TS = E - TS_0 + k_B T \frac{3\mathbf{r}^2}{2DN}. \quad (2.26)$$

This is essentially a thermodynamic version of Hooke's law

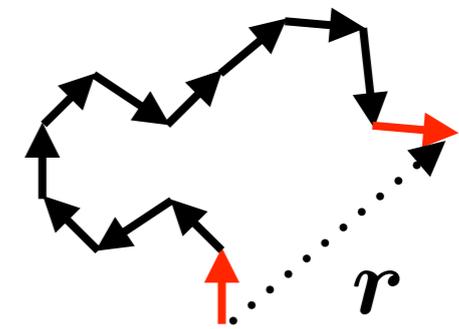
$$F = F_0 + \frac{K}{2} \mathbf{r}^2, \quad K = \frac{3k_B T}{DN}. \quad (2.27)$$

For long stiff polymers we have  $DN \simeq 2\lambda N L_P = 2LL_P$ , we find for the spring-constant

$$K = \frac{3k_B T}{2LL_P}. \quad (2.28)$$

This means, for example, that the persistence length  $L_p$  can be inferred from force measurements if temperature  $T$  and polymer length  $L$  are known.

# Self-avoidance (Flory's scaling argument)

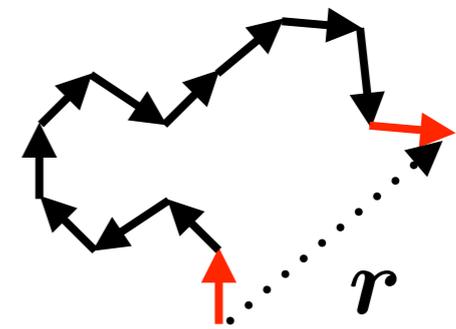


$$F = F_0 + \frac{K}{2} \mathbf{r}^2, \quad K = \frac{3k_B T}{DN}$$

- IDEA: include additional free energy term to account for self repulsion
- ASSUMPTIONS:
  - (i)  $N \gg 1$  monomers of volume  $v_d$  with fixed end-to-end distance  $\mathbf{r}$
  - (ii) for a fixed  $|\mathbf{r}|$ , the  $N$  monomers may (very roughly) explore a volume of  $|\mathbf{r}|^d$ ,
  - (iii) overlap probability given by volume filling fraction  $\phi = v_d N / |\mathbf{r}|^d$ .

$$F_e \simeq Nk_B T \phi = Nk_B T \frac{v_d N}{|\mathbf{r}|^d} \quad \Rightarrow \quad F = F_0 + Nk_B T \left( \frac{v_d N}{|\mathbf{r}|^d} + \frac{|\mathbf{r}|^2 d}{2D_d N^2} \right)$$

# Self-avoidance (Flory's scaling argument)



$$F = F_0 + Nk_B T \left( \frac{v_d N}{|\mathbf{r}|^d} + \frac{|\mathbf{r}|^2 d}{2D_d N^2} \right). \quad (2.30)$$

To obtain the equilibrium distance  $r_*$ , we must minimize this expression with respect to  $r = |\mathbf{r}|$ , which gives

$$0 = \frac{dF}{d|\mathbf{r}|} = -d \frac{v_d N}{r_*^{d+1}} + \frac{d}{D_d N^2} r_* \quad (2.31)$$

and therefore

$$r_* = (D_d v_d)^{1/d+2} N^{3/(d+2)}. \quad (2.32)$$

Thus, explicitly

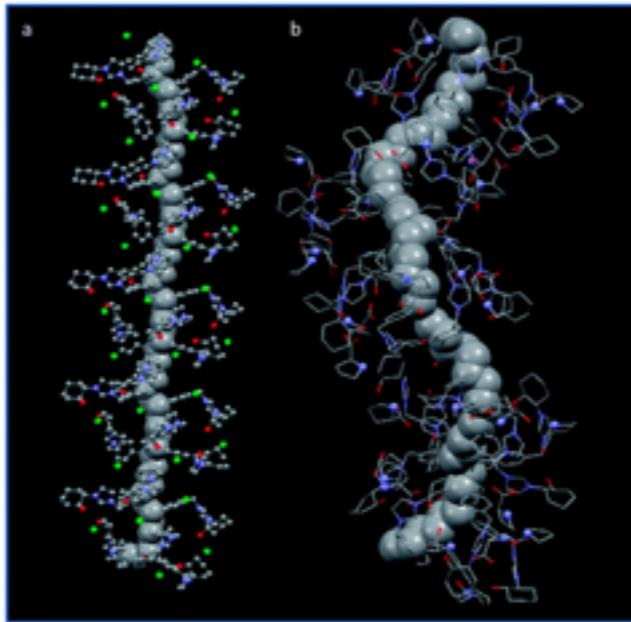
$$d = 1 : \quad r_* \propto N \quad (2.33a)$$

$$d = 2 : \quad r_* \propto N^{3/4}, \quad (2.33b)$$

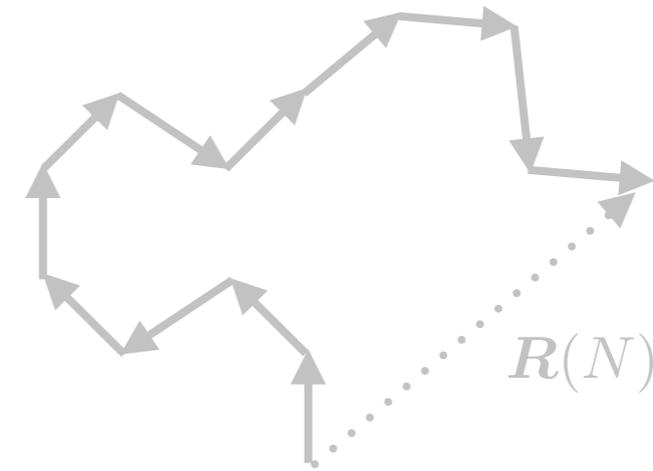
$$d = 3 : \quad r_* \propto N^{3/5}. \quad (2.33c)$$

The result is trivial for  $d = 1$ , seems to be exact for  $d = 2$  when compared to simulations, and is very close to best numerical results  $N^{0.589\dots}$  for  $d = 3$ .

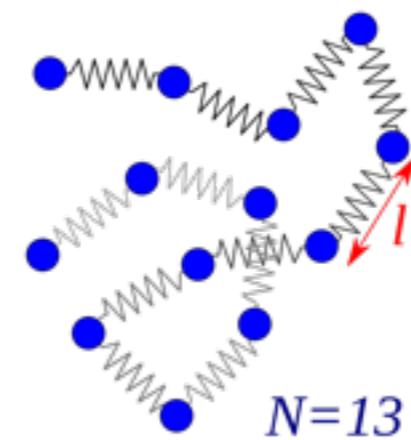
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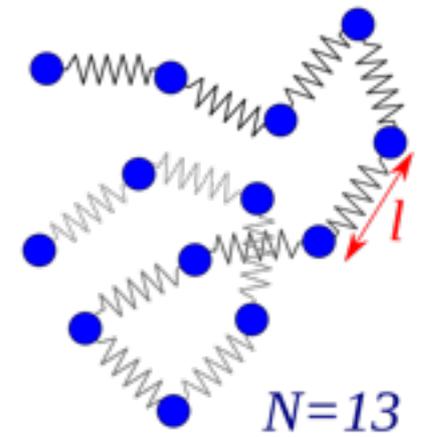


persistent RW model



**Next:** mechanistic model





## 2.2 Bead-spring model

single bead is governed by the over-damped Langevin equation

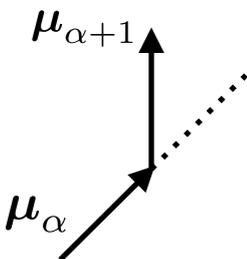
$$d\mathbf{X}_\alpha(t) = -\nabla_{\mathbf{x}_\alpha} U(\{\mathbf{X}_\alpha\}) dt + \sqrt{2D} * d\mathbf{B}_\alpha(t), \quad (2.34)$$

where  $D$  is the thermal diffusion constant of a bead. The potential  $U$  contains contributions from elastic nearest neighbor interactions  $U_e$ , from bending  $U_b$  and, to implement self-avoidance, steric short-range repulsion  $U_s$ :

$$U = U_e + U_b + U_s \quad (2.35)$$

Defining  $(N - 1)$  chain link vectors  $\mathbf{R}_\alpha$  and their orientations  $\boldsymbol{\mu}_\alpha$  by

$$\mathbf{R}_\alpha = \mathbf{X}_{\alpha+1} - \mathbf{X}_\alpha, \quad \boldsymbol{\mu}_\alpha = \frac{\mathbf{R}_\alpha}{\|\mathbf{R}_\alpha\|} \quad (2.36)$$



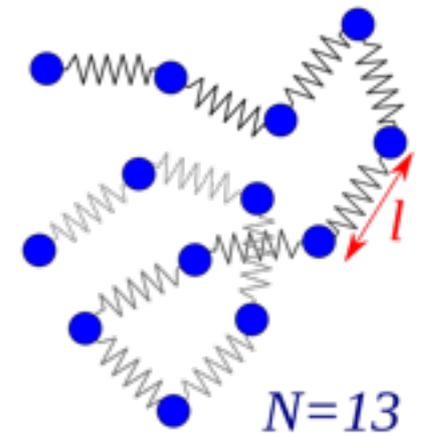
the potentials can be written as sums over 2-body and 3-body interactions

$$U_e = \sum_{\alpha=1}^{N-1} u(\|\mathbf{R}_\alpha\|), \quad U_b = \sum_{\alpha=1}^{N-2} b(\boldsymbol{\mu}_\alpha \cdot \boldsymbol{\mu}_{\alpha+1}), \quad U_s = \sum_{\alpha=1}^N \sum_{\beta=1, \beta \neq \alpha}^N s(\|\mathbf{X}_\alpha - \mathbf{X}_\beta\|).$$

Specifically, the elastic spring potential  $u(r)$  and the steric repulsion potential  $s(r)$  encode 2-body interactions, whereas the bending potential  $b(q)$  involves 3-body interactions.<sup>2</sup>

Plausible choices are

$$u(r) = \frac{K}{2}(r - \lambda)^2, \quad b(q) = \frac{B}{2}(q - 1)^2, \quad s(r) = \frac{S e^{-r/\sigma}}{r^\nu} \quad (2.38)$$



## 2.2 Bead-spring model

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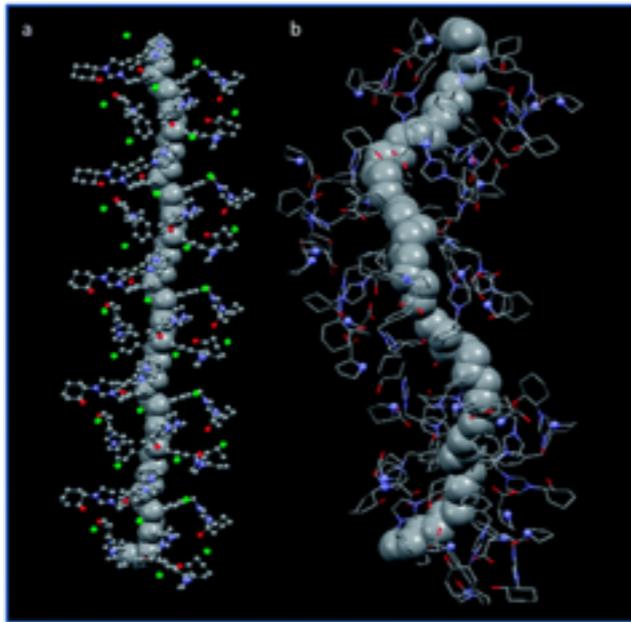
Needs to be solved numerically but **stationary distribution known**

$$p_N(\{\mathbf{x}_\alpha\}) = \frac{1}{Z_N} \exp\left[-\frac{U(\{\mathbf{X}_\alpha\})}{D}\right], \quad (2.39)$$

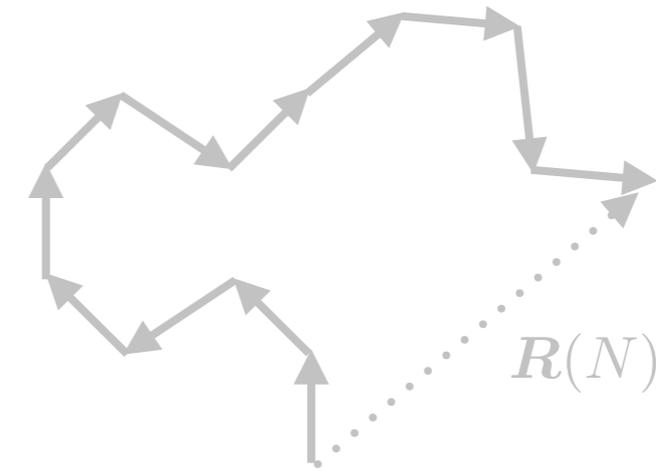
where

$$Z_N = \int \left( \prod_{\alpha=1}^N d^3x_\alpha \right) \exp\left[-\frac{U(\{\mathbf{x}_\alpha\})}{D}\right]. \quad (2.40)$$

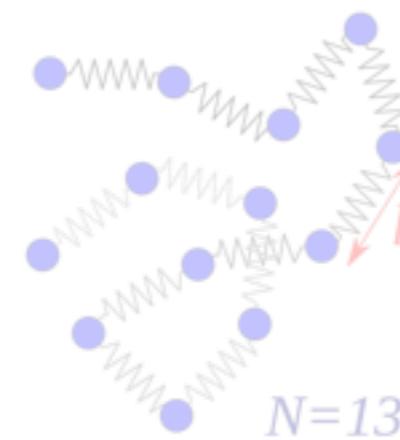
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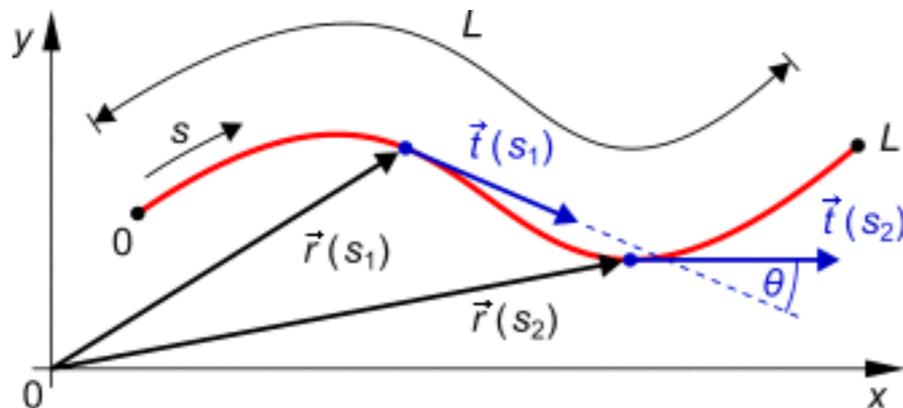
persistent RW model



mechanistic model



continuum model



<http://www.uni-leipzig.de/~pwm/web/?section=introduction&page=polymers>

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## 2.3 Continuum description

### 2.3.1 Differential geometry of curves

Consider a continuous curve  $\mathbf{r}(t) \in \mathbb{R}^3$ , where  $t \in [0, T]$ . Assume that the first three derivatives  $\dot{\mathbf{r}}(t), \ddot{\mathbf{r}}(t), \dddot{\mathbf{r}}(t)$  are linearly independent. The length of the curve is given by

$$L = \int_0^T dt \|\dot{\mathbf{r}}(t)\| \quad (2.41)$$

where  $\dot{\mathbf{r}}(t) = d\mathbf{r}/dt$  and  $\|\cdot\|$  denotes the Euclidean norm. The local unit tangent vector is defined by

$$\mathbf{t} = \frac{\dot{\mathbf{r}}}{\|\dot{\mathbf{r}}\|}. \quad (2.42)$$

The unit normal vector, or unit curvature vector, is

$$\mathbf{n} = \frac{(\mathbf{I} - \mathbf{t}\mathbf{t}) \cdot \ddot{\mathbf{r}}}{\|(\mathbf{I} - \mathbf{t}\mathbf{t}) \cdot \ddot{\mathbf{r}}\|}. \quad (2.43)$$

Unit tangent vector  $\hat{\mathbf{t}}(t)$  and unit normal vector  $\hat{\mathbf{n}}(t)$  span the *osculating* (‘kissing’) plane at point  $t$ . The unit binormal vector is defined by

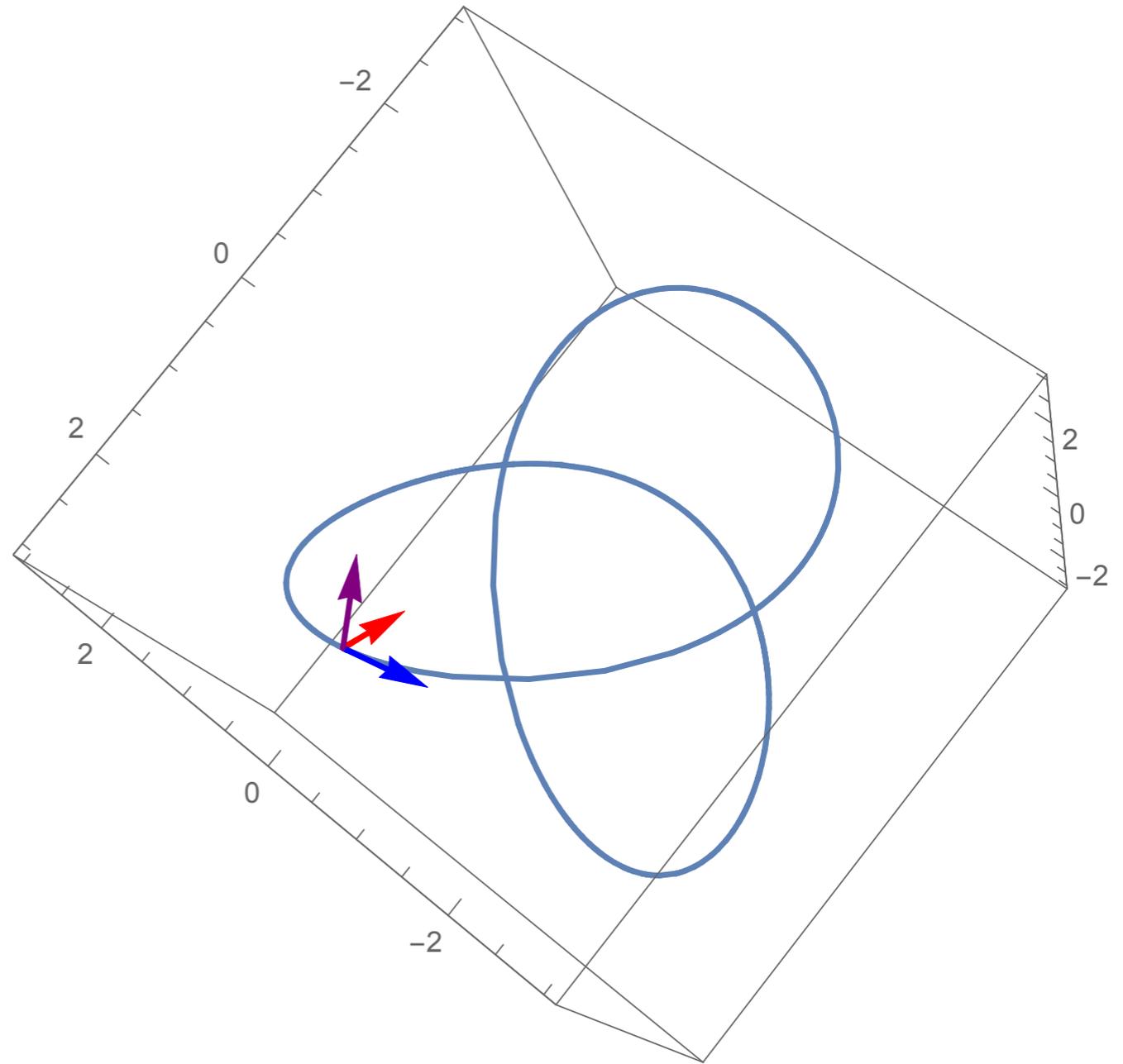
$$\mathbf{b} = \frac{(\mathbf{I} - \mathbf{t}\mathbf{t}) \cdot (\mathbf{I} - \mathbf{n}\mathbf{n}) \cdot \ddot{\mathbf{r}}}{\|(\mathbf{I} - \mathbf{t}\mathbf{t}) \cdot (\mathbf{I} - \mathbf{n}\mathbf{n}) \cdot \ddot{\mathbf{r}}\|}. \quad (2.44)$$

The orthonormal basis  $\{\mathbf{t}(t), \mathbf{n}(t), \mathbf{b}(t)\}$  spans the local Frenet frame. For *plane* curves,  $\ddot{\mathbf{r}}(t)$  is not linearly independent of  $\dot{\mathbf{r}}$  and  $\ddot{\mathbf{r}}$ . In this case, we set  $\mathbf{b} = \mathbf{t} \wedge \mathbf{n}$ .

$$\mathbf{t} = \frac{\dot{\mathbf{r}}}{\|\dot{\mathbf{r}}\|}$$

$$\mathbf{n} = \frac{(\mathbf{I} - \mathbf{t}\mathbf{t}) \cdot \ddot{\mathbf{r}}}{\|(\mathbf{I} - \mathbf{t}\mathbf{t}) \cdot \ddot{\mathbf{r}}\|}$$

$$\mathbf{b} = \frac{(\mathbf{I} - \mathbf{t}\mathbf{t}) \cdot (\mathbf{I} - \mathbf{n}\mathbf{n}) \cdot \ddot{\mathbf{r}}}{\|(\mathbf{I} - \mathbf{t}\mathbf{t}) \cdot (\mathbf{I} - \mathbf{n}\mathbf{n}) \cdot \ddot{\mathbf{r}}\|}$$



The local curvature  $\kappa(t)$  and the associated radius of curvature  $\rho(t) = 1/\kappa$  are defined by

$$\kappa(t) = \frac{\dot{\mathbf{t}} \cdot \mathbf{n}}{\|\dot{\mathbf{r}}\|}, \quad (2.45)$$

and the local torsion  $\tau(t)$  by

$$\tau(t) = \frac{\dot{\mathbf{n}} \cdot \mathbf{b}}{\|\dot{\mathbf{r}}\|}. \quad (2.46)$$

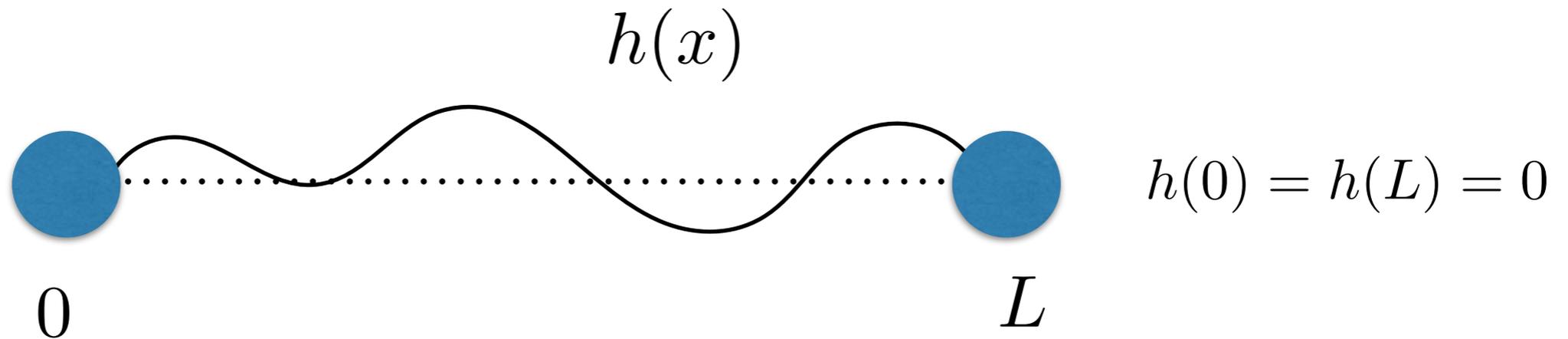
For plane curves with constant  $\mathbf{b}$ , we have  $\tau = 0$ .

Given  $\|\dot{\mathbf{r}}\|$ ,  $\kappa(t)$ ,  $\tau(t)$  and the initial values  $\{\mathbf{t}(0), \mathbf{n}(0), \mathbf{b}(0)\}$ , the Frenet frames along the curve can be obtained by solving the Frenet-Serret system

$$\frac{1}{\|\dot{\mathbf{r}}\|} \begin{pmatrix} \dot{\mathbf{t}} \\ \dot{\mathbf{n}} \\ \dot{\mathbf{b}} \end{pmatrix} = \begin{pmatrix} 0 & \kappa & 0 \\ -\kappa & 0 & \tau \\ 0 & -\tau & 0 \end{pmatrix} \begin{pmatrix} \mathbf{t} \\ \mathbf{n} \\ \mathbf{b} \end{pmatrix}. \quad (2.47a)$$

The above formulas simplify if  $t$  is the arc length, for in this case  $\|\dot{\mathbf{r}}\| = 1$ .

### 2.3.2 Stretchable polymers: Minimal model and equipartition



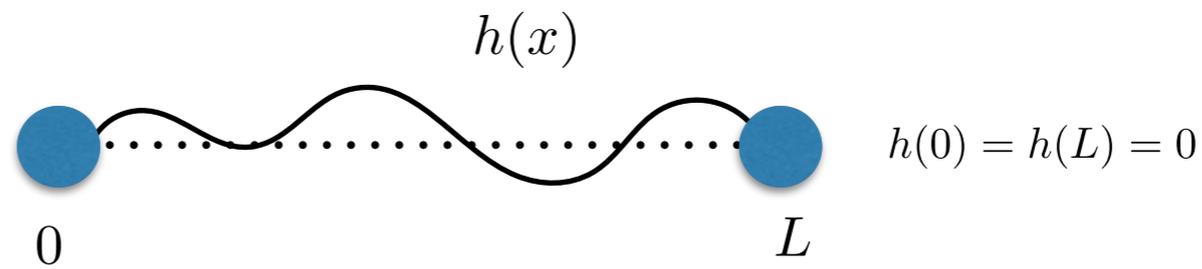
$$E = \gamma \left[ \int_0^L dx \sqrt{1 + h_x^2} - L \right], \quad (2.48)$$

where  $h_x = h'(x)$ . Restricting ourselves to small deformations,  $|h_x| \ll 1$ , we may approximate

$$E \simeq \frac{\gamma}{2} \int_0^L dx h_x^2. \quad (2.49)$$

---

<sup>3</sup> $\gamma$  carries units of energy/length.



$$E \simeq \frac{\gamma}{2} \int_0^L dx h_x^2. \quad (2.49)$$

Taking into account that  $h(0) = h(L) = 0$ , we may represent  $h(x)$  and its derivative through the Fourier-sine series

$$h(x) = \sum_{n=1}^{\infty} A_n \sin\left(\frac{n\pi x}{L}\right) \quad (2.50a)$$

$$h_x(x) = \sum_{n=1}^{\infty} A_n \frac{n\pi}{L} \cos\left(\frac{n\pi x}{L}\right). \quad (2.50b)$$

Exploiting orthogonality

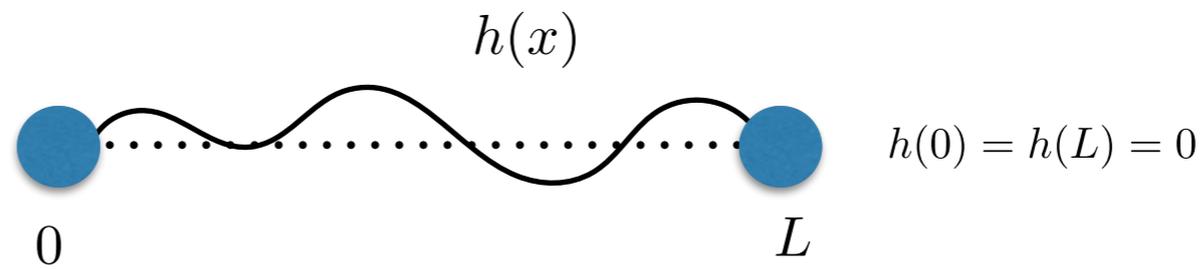
$$\int_0^L dx \sin\left(\frac{n\pi x}{L}\right) \sin\left(\frac{m\pi x}{L}\right) = \frac{L}{2} \delta_{nm} \quad (2.51)$$

we may rewrite the energy (2.49) as

$$\begin{aligned} E &\simeq \frac{\gamma}{2} \sum_n \sum_m \int_0^L dx A_n A_m \left(\frac{n\pi}{L}\right) \left(\frac{m\pi}{L}\right) \cos\left(\frac{n\pi x}{L}\right) \cos\left(\frac{m\pi x}{L}\right) \\ &= \frac{\gamma}{2} \sum_n \sum_m A_n A_m \left(\frac{n\pi}{L}\right) \left(\frac{m\pi}{L}\right) \frac{L}{2} \delta_{nm} \\ &= \sum_{n=1}^{\infty} E_n, \end{aligned} \quad (2.52a)$$

where the energy  $E_n$  stored in Fourier mode  $n$  is

$$E_n = A_n^2 \left(\frac{\gamma n^2 \pi^2}{4L}\right). \quad (2.52b)$$



$$E \simeq \frac{\gamma}{2} \int_0^L dx h_x^2 = \sum_{n=1}^{\infty} E_n \qquad E_n = A_n^2 \left( \frac{\gamma n^2 \pi^2}{4L} \right)$$

Now assume the polymer is coupled to a bath and the stationary distribution is canonical

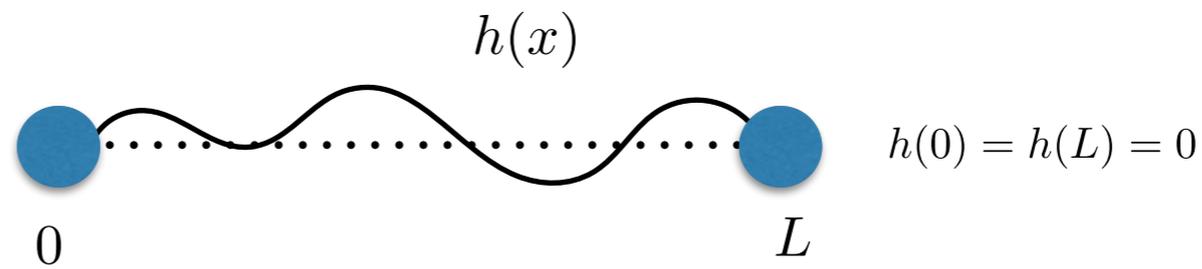
$$\begin{aligned} p(\{A_n\}) &= \frac{1}{Z} \exp(-\beta E) \\ &= \frac{1}{Z} \exp \left[ -\beta \sum_{n=1}^{\infty} A_n^2 \left( \frac{\gamma n^2 \pi^2}{4L} \right) \right] \end{aligned} \qquad (2.53)$$

with  $\beta = (k_B T)^{-1}$ . The PDF factorizes and, therefore, also the normalization constant

$$Z = \prod_{i=1}^{\infty} Z_n, \qquad (2.54a)$$

where

$$Z_n = \int_{-\infty}^{\infty} dA_n \exp \left[ -\beta A_n^2 \left( \frac{\gamma n^2 \pi^2}{4L} \right) \right] = \left( \frac{4\pi L}{\beta \gamma n^2 \pi^2} \right)^{1/2}. \qquad (2.54b)$$



$$E \simeq \frac{\gamma}{2} \int_0^L dx h_x^2 = \sum_{n=1}^{\infty} E_n \quad E_n = A_n^2 \left( \frac{\gamma n^2 \pi^2}{4L} \right)$$

Now assume the polymer is coupled to a bath and the stationary distribution is canonical

$$\begin{aligned} p(\{A_n\}) &= \frac{1}{Z} \exp(-\beta E) \\ &= \frac{1}{Z} \exp \left[ -\beta \sum_{n=1}^{\infty} A_n^2 \left( \frac{\gamma n^2 \pi^2}{4L} \right) \right] \end{aligned} \quad (2.53)$$

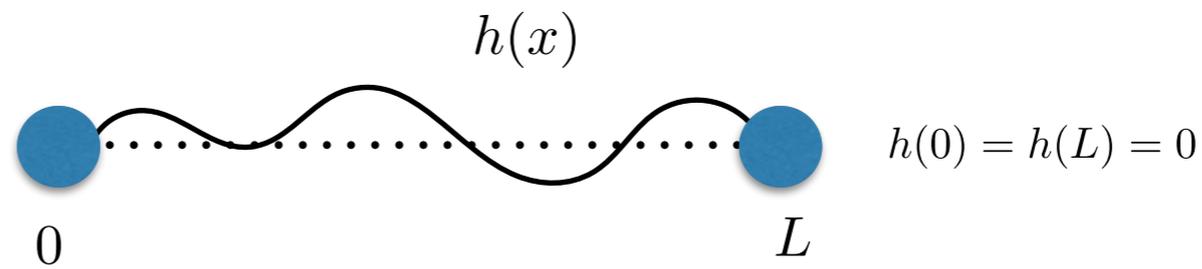
We thus find for the first to moments of  $A_n$

$$\mathbb{E}[A_n] = 0 \quad (2.55a)$$

$$\mathbb{E}[A_n^2] = \frac{2k_B T L}{\gamma n^2 \pi^2}, \quad (2.55b)$$

and from this for the mean energy per mode

$$\mathbb{E}[E_n] = \left( \frac{\gamma n^2 \pi^2}{4L} \right) \mathbb{E}[A_n^2] = \frac{1}{2} k_B T. \quad (2.56)$$



We may use the equipartition result to compute the variance of the polymer at the position  $x \in [0, L]$

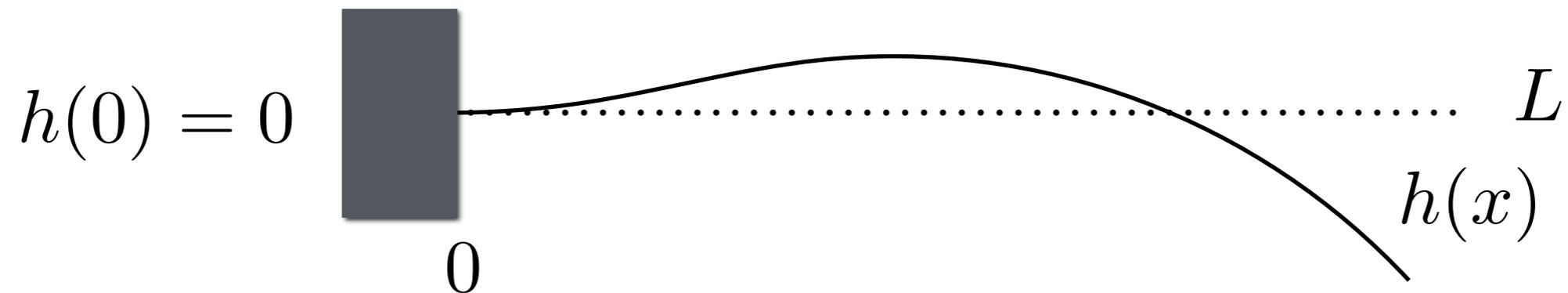
$$\begin{aligned}
 \mathbb{E}[h(x)^2] &:= \sum_{n=1}^{\infty} \sum_{m=1}^{\infty} \mathbb{E}[A_n A_m] \sin\left(\frac{n\pi x}{L}\right) \sin\left(\frac{m\pi x}{L}\right) \\
 &= \sum_{n=1}^{\infty} \sum_{m=1}^{\infty} \mathbb{E}[A_n^2] \delta_{nm} \sin\left(\frac{n\pi x}{L}\right) \sin\left(\frac{m\pi x}{L}\right) \\
 &= \left(\frac{2k_B T L}{\gamma \pi^2}\right) \sum_{n=1}^{\infty} \frac{\sin^2(n\pi x/L)}{n^2}.
 \end{aligned} \tag{2.57}$$

If we additionally average along  $x$

$$\langle \mathbb{E}[h(x)^2] \rangle = \left(\frac{k_B T L}{\gamma \pi^2}\right) \sum_{n=1}^{\infty} \frac{1}{n^2} = \left(\frac{k_B T L}{\gamma \pi^2}\right) \frac{\pi^2}{6} = \frac{k_B T L}{6\gamma}. \tag{2.58}$$

Thus, by measuring fluctuations along the polymer we may infer  $\gamma$ .

### 2.3.3 Rigid polymers: Euler-Bernoulli equation



$$E \simeq \frac{A}{2} \int_0^L dx \kappa^2, \quad (2.59)$$

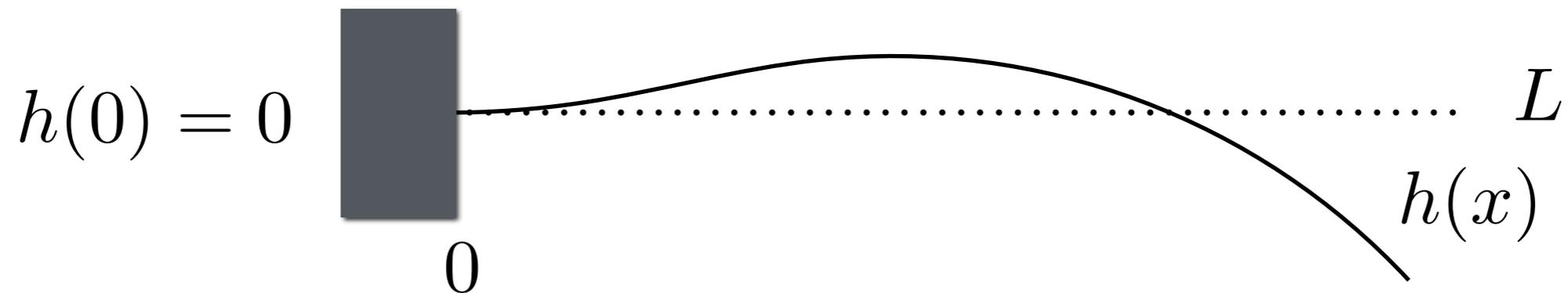
where  $A$  is the bending modulus (units energy  $\times$  length). For plane curves  $h(x)$ , the curvature can be expressed as

$$\kappa = \frac{h_{xx}}{(1 + h_x^2)^{3/2}}. \quad (2.60)$$

Focussing on the limit of weak deformations,  $h_x \ll 1$ , we may approximate  $\kappa \simeq h_{xx}$ , and the energy simplifies to

$$E \simeq \frac{A}{2} \int_0^L dx (h_{xx})^2. \quad (2.61)$$

# Boundary conditions



$$E \simeq \frac{A}{2} \int_0^L dx (h_{xx})^2. \quad (2.61)$$

The exact form of the boundary conditions depend on how the polymer is attached to the plane  $x = 0$ . Assuming that polymer is rigidly anchored at an angle  $90^\circ$ , the boundary conditions at the fixed end at  $x = 0$  are

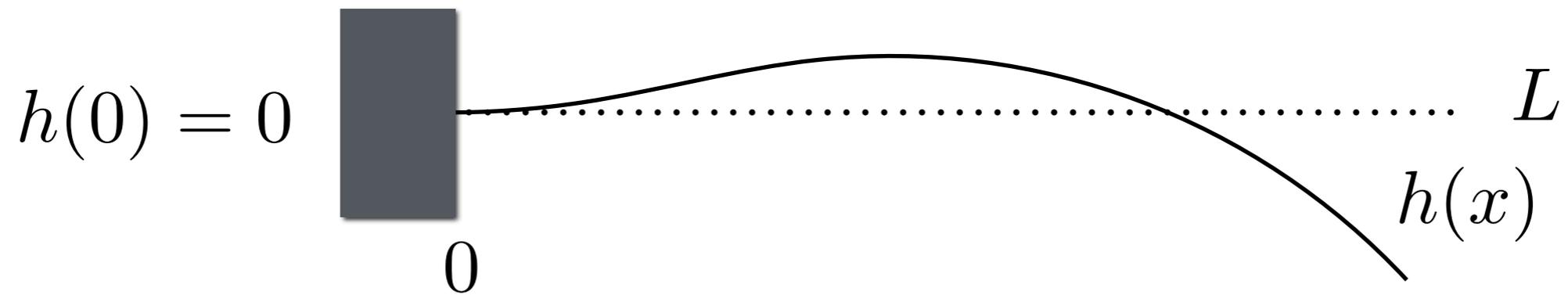
$$h(0) = 0, \quad h_x(0) = 0. \quad (2.62a)$$

At the free end, we will consider flux conditions

$$h_{xx}(L) = 0, \quad h_{xxx}(L) = 0. \quad (2.62b)$$

(minimal absolute curvature at the free end)

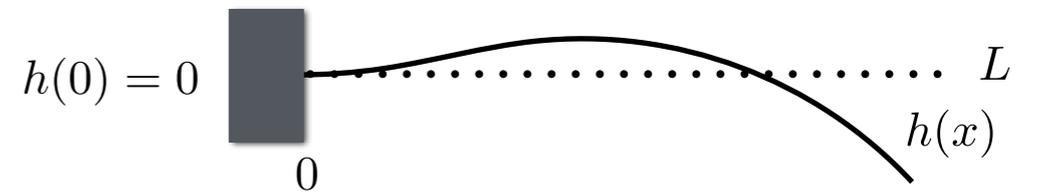
# Boundary conditions



$$E \simeq \frac{A}{2} \int_0^L dx (h_{xx})^2. \quad (2.61)$$

By partial integrations, we may rewrite (2.61) as

$$\begin{aligned} E &\simeq \frac{A}{2} \left[ h_x h_{xx} \Big|_0^L - \int_0^L dx h_x h_{xxx} \right] \\ &= \frac{A}{2} \left[ - \int_0^L dx h_x h_{xxx} \right] \\ &= \frac{A}{2} \left[ -h h_{xxx} \Big|_0^L + \int_0^L dx h h_{xxxx} \right] = \frac{A}{2} \left[ \int_0^L dx h h_{xxxx} \right]. \end{aligned} \quad (2.63)$$



If the polymer is surrounded by a viscous solvent, an initial perturbation  $h(0, x)$  will relax to the ground-state. Neglecting fluctuations due to thermal noise, the relaxation dynamics  $h(t, x)$  will be of the over-damped form<sup>4</sup>

$$\eta h_t = -\frac{\delta E}{\delta h}, \quad (2.64)$$

where  $\eta$  is a damping constant, and the variational derivative is defined by

$$\frac{\delta E[h(x)]}{\delta h(y)} := \lim_{\epsilon \rightarrow 0} \frac{E[h(x) + \epsilon \delta(x - y)] - E[h(x)]}{\epsilon}. \quad (2.65)$$

Keeping terms up to order  $\epsilon$ , we find for the energy functional (2.61)

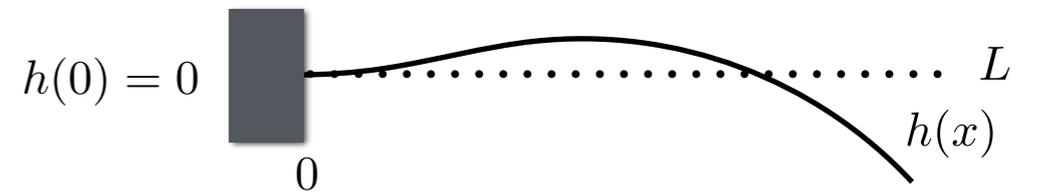
$$\begin{aligned} E[h(x) + \epsilon \delta(x - y)] - E[h(x)] &= \frac{A}{2} \int_0^L dx [(h + \epsilon \delta)_{xx} (h + \epsilon \delta)_{xx} - (h_{xx})^2] \\ &= \frac{A}{2} \int_0^L dx [2\epsilon h_{xx} \delta_{xx} + \mathcal{O}(\epsilon^2)] \end{aligned}$$

Using the integral identity

$$g(x) \partial_x^n \delta(x - y) = (-1)^n \delta(x - y) \partial_x^n g(x) \quad (2.66)$$

for any smooth function  $g$ , one obtains

$$\frac{\delta E[h(x)]}{\delta h(y)} = A \int_0^L dx h_{xxxx}(x) \delta(x - y) = Ah_{xxxx}(y), \quad (2.67)$$



If the polymer is surrounded by a viscous solvent, an initial perturbation  $h(0, x)$  will relax to the ground-state. Neglecting fluctuations due to thermal noise, the relaxation dynamics  $h(t, x)$  will be of the over-damped form<sup>4</sup>

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so that Eq. (2.64) becomes a linear fourth-order equation

$$h_t = -\alpha h_{xxxx}, \quad \alpha = \frac{A}{\eta}. \quad (2.68)$$

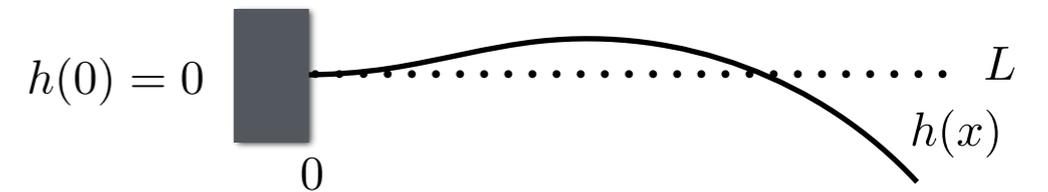
Inserting the ansatz

$$h = e^{-t/\tau} \phi(x), \quad h_t = -\frac{1}{\tau} e^{-t/\tau} \phi, \quad h_{xxxx} = e^{-t/\tau} \phi_{xxxx}, \quad (2.69)$$

gives the eigenvalue problem

$$\frac{1}{\tau \alpha} \phi = \phi_{xxxx}. \quad (2.70)$$

# Eigenvalue problem



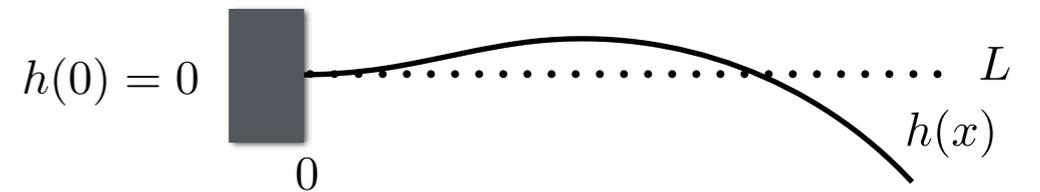
$$\frac{1}{\tau\alpha}\phi = \phi_{xxxx}. \quad (2.70)$$

for the one-dimensional biharmonic operator  $(\partial_x^2)^2$ , which has the general solution

$$\phi(x) = B_1 \cosh(x/\lambda) + B_2 \sinh(x/\lambda) + B_3 \cos(x/\lambda) + B_4 \sin(x/\lambda) \quad (2.71a)$$

where

$$\lambda = (\alpha\tau)^{1/4}. \quad (2.71b)$$



Inserting the first two conditions into the last two, we obtain the linear system

$$0 = B_1[\cosh(L/\lambda) + \cos(L/\lambda)] + B_2[\sinh(L/\lambda) + \sin(L/\lambda)] \quad (2.73a)$$

$$0 = B_1[\sinh(L/\lambda) - \sin(L/\lambda)] + B_2[\cosh(L/\lambda) + \cos(L/\lambda)]. \quad (2.73b)$$

For nontrivial solutions to exist, we must have

$$0 = \det \begin{pmatrix} [\cosh(L/\lambda) + \cos(L/\lambda)] & [\sinh(L/\lambda) + \sin(L/\lambda)] \\ [\sinh(L/\lambda) - \sin(L/\lambda)] & [\cosh(L/\lambda) + \cos(L/\lambda)] \end{pmatrix} \quad (2.74)$$

which gives us the eigenvalue condition

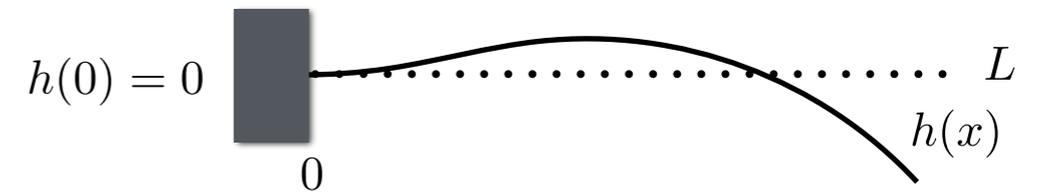
$$0 = \cosh(L/\lambda) \cos(L/\lambda) + 1. \quad (2.75)$$

This equation has solutions for discrete values  $\lambda_n > 0$  that can be computed numerically, and one finds for the first few eigenvalues

$$\frac{L}{2\lambda_n} = \{0.94, 2.35, 3.93, 5.50, \dots\}. \quad (2.76)$$

For comparison, for purely sinusoidal excitations of a harmonic string

$$L/\lambda_n \propto n.$$

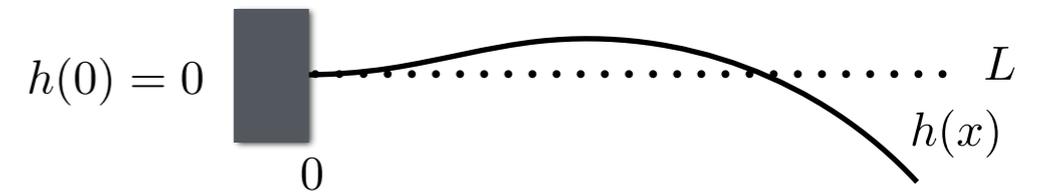


The full time-dependent solution can thus be written as

$$h(t, x) = \sum_{n=1}^{\infty} B_{1n} e^{-t/\tau_n} \left\{ \cosh(x/\lambda_n) - \cos(x/\lambda_n) + \frac{\cos(L/\lambda_n) + \cosh(L/\lambda_n)}{\sin(L/\lambda_n) + \sinh(L/\lambda_n)} [\sin(x/\lambda_n) - \sinh(x/\lambda_n)] \right\}, \quad (2.77)$$

Limit  $\eta \rightarrow \infty$  or  $t \rightarrow 0$

$$h(x) = \sum_{n=1}^{\infty} B_{1n} \left\{ \cosh(x/\lambda_n) - \cos(x/\lambda_n) + \frac{\cos(L/\lambda_n) + \cosh(L/\lambda_n)}{\sin(L/\lambda_n) + \sinh(L/\lambda_n)} [\sin(x/\lambda_n) - \sinh(x/\lambda_n)] \right\}$$

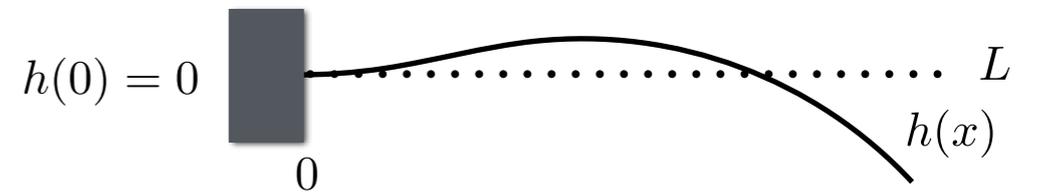


The full time-dependent solution can thus be written as

$$h(t, x) = \sum_{n=1}^{\infty} B_{1n} e^{-t/\tau_n} \left\{ \cosh(x/\lambda_n) - \cos(x/\lambda_n) + \frac{\cos(L/\lambda_n) + \cosh(L/\lambda_n)}{\sin(L/\lambda_n) + \sinh(L/\lambda_n)} [\sin(x/\lambda_n) - \sinh(x/\lambda_n)] \right\}, \quad (2.77)$$

Limit  $\eta \rightarrow \infty$  or  $t \rightarrow 0$

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$$h(x) = \sum_{n=1}^{\infty} B_{1n} \left\{ \cosh(x/\lambda_n) - \cos(x/\lambda_n) + \frac{\cos(L/\lambda_n) + \cosh(L/\lambda_n)}{\sin(L/\lambda_n) + \sinh(L/\lambda_n)} [\sin(x/\lambda_n) - \sinh(x/\lambda_n)] \right\}$$

This expression can be inserted into (2.63), and after exploiting orthogonality of the bi-harmonic eigenfunctions

$$E \simeq \sum_{n=1} E_n, \quad E_n = \frac{A}{2} \frac{L}{\lambda_n^4} B_n^2, \quad (2.79)$$

i.e., the energy per mode is proportional to the square of the amplitude, just as in the stretching case discussed in Sec. 2.3.2. It is therefore possible to compute thermal expectation values exactly from Gaussian integrals. In particular, from equipartition

# Actin in flow

PRL 108, 038103 (2012)

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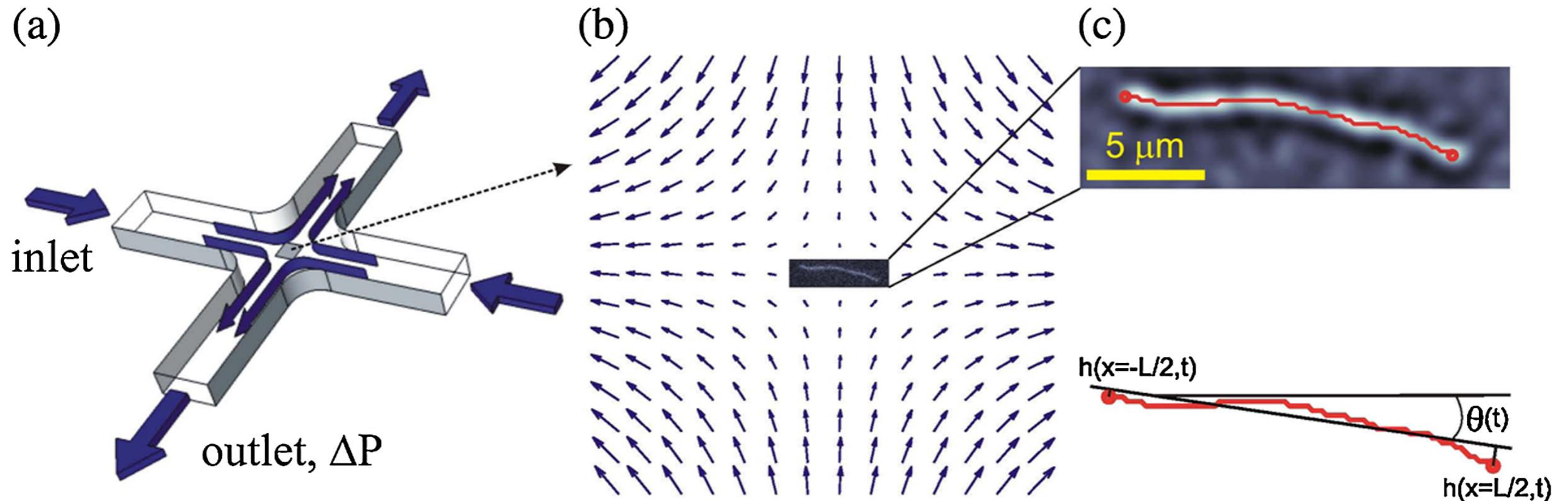
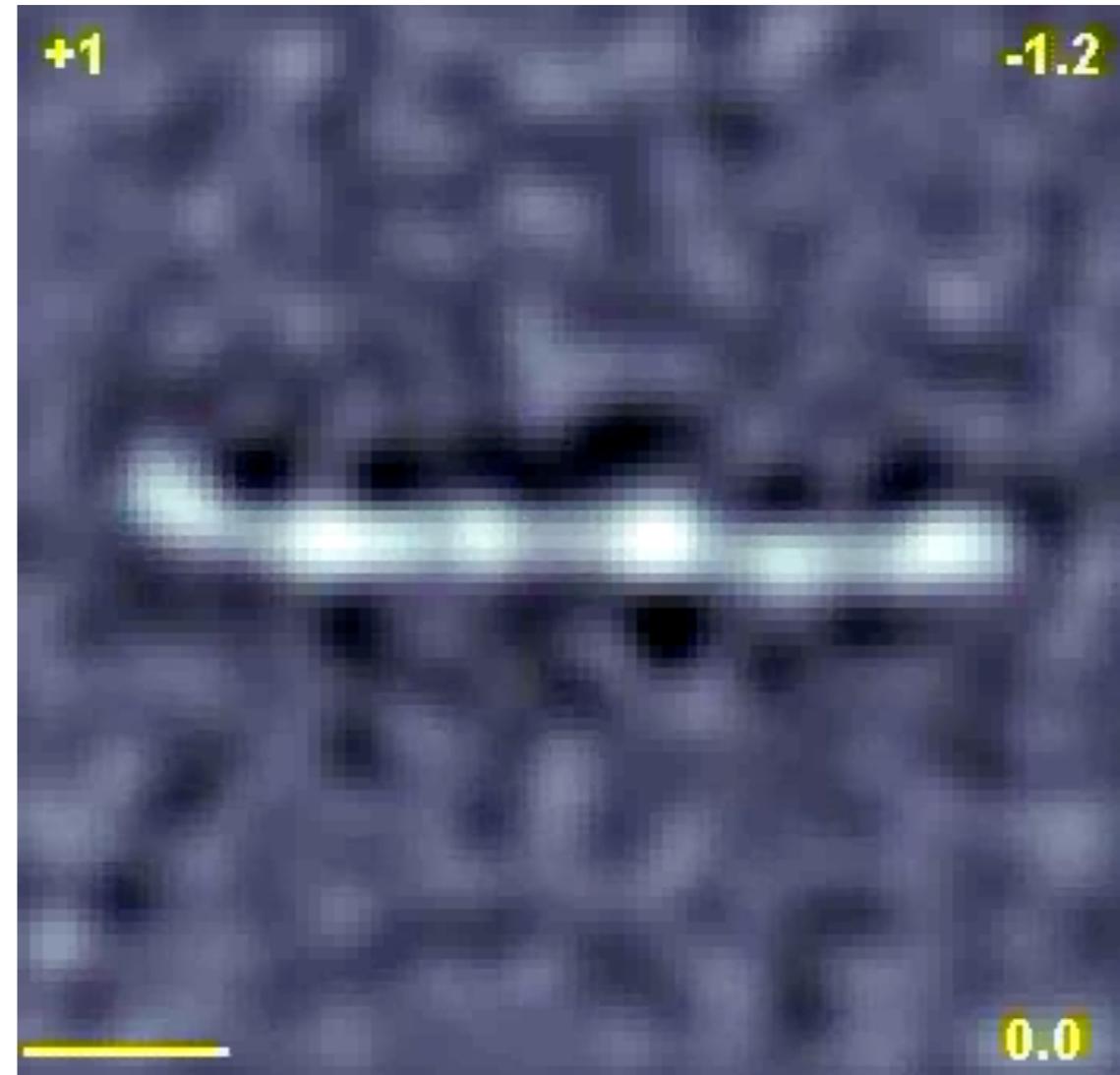
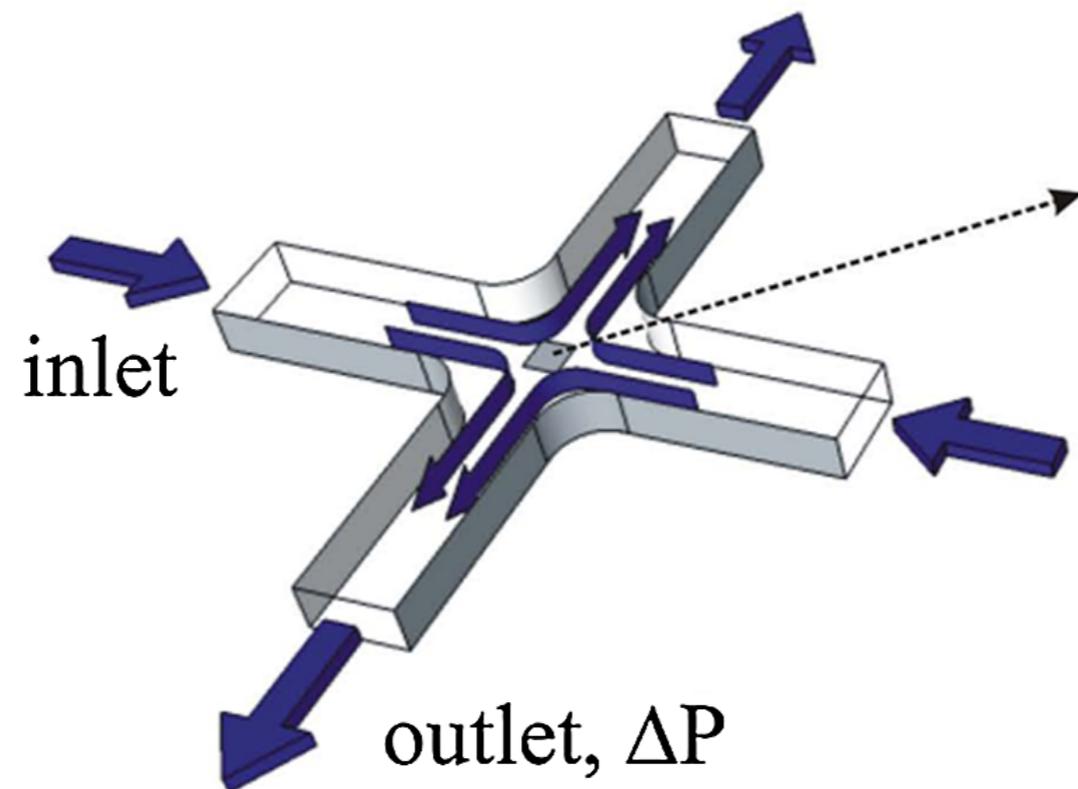


FIG. 1 (color online). Experimental setup. (a) Microfluidic cross-flow geometry controlled by a pressure difference  $\Delta P$  between inlet and outlet branches. (b) Close-up of the velocity field near the stagnation point, showing a typical actin filament. (c) Raw contour (red) of an actin filament and definition of geometric quantities used in the analysis.

Kantsler & Goldstein (2012) PRL

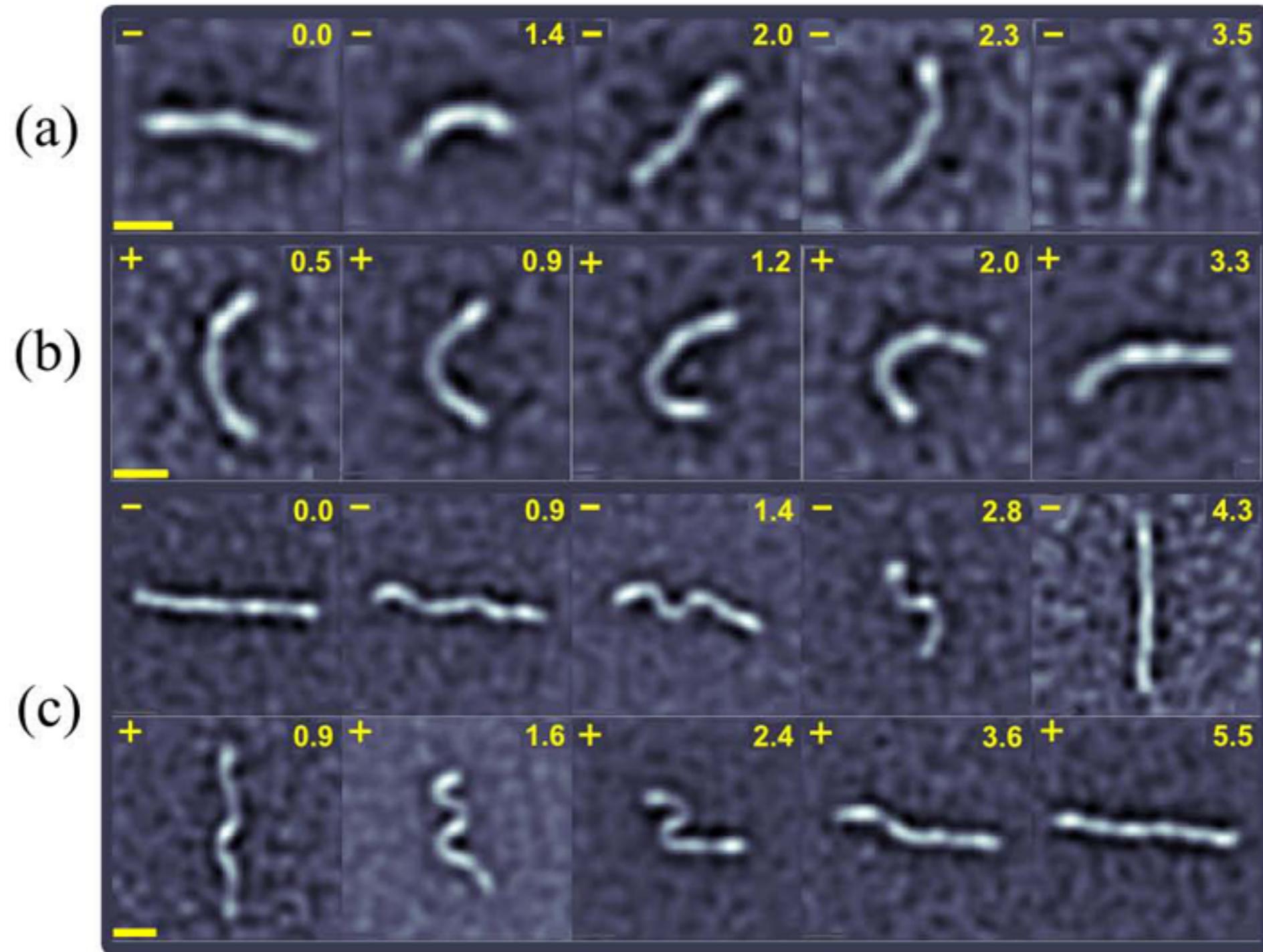
# Actin in flow

(a)



Kantsler & Goldstein (2012) PRL

# Actin in flow



Kantsler & Goldstein (2012) PRL

# Theory

$$\mathcal{E} = \frac{1}{2} \int_{-L/2}^{L/2} dx \{ Ah_{xx}^2 + \sigma(x)h_x^2 \}, \quad (1)$$

where subscripts indicate differentiation. The *nonuniform* tension induced by the flow [19],

$$\sigma(x) = \frac{2\pi\mu\dot{\gamma}}{\ln(1/\epsilon^2 e)} (L^2/4 - x^2), \quad (2)$$

# Theory

of eigenfunctions  $W^{(n)}$  (and eigenvalues  $\lambda_n$ ) with boundary conditions  $W_{xx}(\pm L/2) = W_{xxx}(\pm L/2) = 0$  [3,21]. Under the convenient rescaling  $\xi = \pi x/L$ , these obey

$$W_{4\xi}^{(n)} - \Sigma \partial_{\xi} [(\pi^2/4 - \xi^2) W_{\xi}^{(n)}] = \Lambda_n W^{(n)}. \quad (3)$$

The eigenvalues  $\Lambda_n = L^4 \lambda_n / \pi^4 A$  are functions of [22]

$$\Sigma = \frac{2\mu \dot{\gamma} L^4}{\pi^3 A \ln(1/\epsilon^2 e)}. \quad (4)$$

When  $\Sigma = 0$ , the  $W^{(n)}$  are eigenfunctions of the one-dimensional biharmonic equation

$$W_{\Sigma=0} = A \sin kx + B \sinh kx + D \cos kx + E \cosh kx. \quad (5)$$

# Theory vs. experiment

(and we assume they are normalized). Equipartition then yields  $\langle a_m a_n \rangle = \delta_{mn} L^4 / \pi^4 \ell_p \Lambda_n$ , and the local variance  $V(x) = \langle [h(x) - \bar{h}]^2 \rangle$  is

$$V(x; \Sigma) = \frac{L^3}{\ell_p \pi^4} \sum_{n=1}^{\infty} \frac{W^{(n)}(x)^2}{\Lambda_n(\Sigma)}.$$

